



Cite this: DOI: 10.1039/c8na00314a

# A comparative study of mechanisms of the adsorption of CO<sub>2</sub> confined within graphene–MoS<sub>2</sub> nanosheets: a DFT trend study†

Francis M. Enejekwu,<sup>ID</sup><sup>ab</sup> Collins I. Ezeh,<sup>ID</sup><sup>a</sup> Michael W. George,<sup>ac</sup> Mengxia Xu,<sup>ID</sup><sup>ab</sup> Hainam Do,<sup>ab</sup> Yue Zhang,<sup>ID</sup><sup>d</sup> Haitao Zhao<sup>ID</sup><sup>b</sup> and Tao Wu<sup>ID</sup><sup>\*ab</sup>

The space within the interlayer of 2-dimensional (2D) nanosheets provides new and intriguing confined environments for molecular interactions. However, atomic level understanding of the adsorption mechanism of CO<sub>2</sub> confined within the interlayer of 2D nanosheets is still limited. Herein, we present a comparative study of the adsorption mechanisms of CO<sub>2</sub> confined within graphene–molybdenum disulfide (MoS<sub>2</sub>) nanosheets using density functional theory (DFT). A comprehensive analysis of CO<sub>2</sub> adsorption energies ( $E_{AE}$ ) at various interlayer spacings of different multilayer structures comprising graphene/graphene (GrapheneB) and MoS<sub>2</sub>/MoS<sub>2</sub> (MoS<sub>2</sub>B) bilayers as well as graphene/MoS<sub>2</sub> (GMoS<sub>2</sub>) and MoS<sub>2</sub>/graphene (MoS<sub>2</sub>G) hybrids is performed to obtain the most stable adsorption configurations. It was found that 7.5 Å and 8.5 Å interlayer spacings are the most stable conformations for CO<sub>2</sub> adsorption on the bilayer and hybrid structures, respectively. Adsorption energies of the multilayer structures decreased in the following trend: MoS<sub>2</sub>B > GrapheneB > MoS<sub>2</sub>G > GMoS<sub>2</sub>. By incorporating van der Waals (vdW) interactions between the CO<sub>2</sub> molecule and the surfaces, we find that CO<sub>2</sub> binds more strongly on these multilayer structures. Furthermore, there is a slight discrepancy in the binding energies of CO<sub>2</sub> adsorption on the heterostructures (GMoS<sub>2</sub>, MoS<sub>2</sub>G) due to the modality of the atom arrangement (C–Mo–S–O and Mo–S–O–C) in both structures, indicating that conformational anisotropy determines to a certain degree its CO<sub>2</sub> adsorption energy. Meanwhile, Bader charge analysis shows that the interaction between CO<sub>2</sub> and these surfaces causes charge transfer and redistributions. By contrast, the density of states (DOS) plots show that CO<sub>2</sub> physisorption does not have a substantial effect on the electronic properties of graphene and MoS<sub>2</sub>. In summary, the results obtained in this study could serve as useful guidance in the preparation of graphene–MoS<sub>2</sub> nanosheets for the improved adsorption efficiency of CO<sub>2</sub>.

Received 28th October 2018  
Accepted 4th January 2019

DOI: 10.1039/c8na00314a

rsc.li/nanoscale-advances

## 1 Introduction

Identifying stable systems with improved CO<sub>2</sub> capture has become an essential goal for carbon capture and storage technologies.<sup>1,2</sup> In recent years, gas-adsorbent materials (GAMs) have received significant attention from many researchers across the globe due to the increased rate of effluent gases (for example, CO<sub>x</sub>, NO<sub>x</sub>, SO<sub>x</sub>, Hg<sup>0</sup>, *etc.*) from industries.<sup>3</sup> Among all these, carbon dioxide (CO<sub>2</sub>)

concentration in the atmosphere has been found to be increasing rapidly (313 ppm, 1960 to 407 ppm, 2017),<sup>4</sup> and this has been linked to the increased reliance on fossil fuels within the past century. Accordingly, reduction and capture of greenhouse gases have been among the most challenging issues in environmental protection.<sup>5</sup> In the past decades, the use of capturing and separation techniques like absorption, adsorption, use of membranes and the ilk was widespread, however, the efficiencies of these processes seem to pose a serious challenge.<sup>6</sup> In the bid to tackle the issue of global warming, energy-efficient capture of CO<sub>2</sub> and other greenhouse gases through industrially proven and simple processes like adsorption is of critical importance. Although a large number of gas-adsorbent materials have been computationally and experimentally developed in the past, nanostructured GAMs are at the cutting-edge of potentially revolutionary advancements in fast growing technological fields such as molecular sensing, energy storage and harvesting, environmental and sustainability engineering, *etc.*<sup>1–3,7,8</sup> This has led to

<sup>a</sup>Department of Chemical and Environmental Engineering, University of Nottingham Ningbo China, Ningbo 315100, China. E-mail: tao.wu@nottingham.ac.uk; tao.wu@hotmail.com

<sup>b</sup>New Material Institute, University of Nottingham Ningbo China, Ningbo 315042, China

<sup>c</sup>School of Chemistry, University of Nottingham, Nottingham NG72RD, UK

<sup>d</sup>Institute of Process Engineering, Chinese Academy of Sciences, Beijing 100190, China

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c8na00314a



the rapid increase in the development of novel nanostructured GAMs. Their properties, which include high surface to volume ratio, regular atomic composition, tunable reactivity, effective transport properties and assembling affinity to form supra-molecular systems, have warranted their applications in CO<sub>2</sub> capture technologies.<sup>3</sup>

Among these nanostructured materials, two-dimensional (2D) nanomaterials like graphene and hexagonal-BN (hBN) sheets have attracted greater interest because of their extraordinary properties attributed to their ultrathin thickness, which is related to quantum effects.<sup>9–11</sup> These 2D nanomaterials vary in terms of their electronic properties and functionality. For instance, graphene is a low-work function metallic electrode, hBN is an insulator, and MoS<sub>2</sub> is generally an n-type semiconductor (very active potential site for CO<sub>2</sub> capture). However, their synergetic function is more attractive for technological application. Recently, several new classes of 2D nanostructures like transition metal dichalcogenides (TMDs) have been studied with more emphasis on molybdenum disulfide (MoS<sub>2</sub>), which exhibits a number of interesting properties such as high catalytic activity,<sup>12</sup> bandgap variation with number of layers,<sup>13,14</sup> high carrier mobility<sup>15,16</sup> and gas sensing capability that have made it widely used in the adsorption of gas molecules especially toxic gases.<sup>17–19</sup> In parallel with the efforts on graphene-like materials, recent studies have focused on the advantage of merging individual properties of different 2D materials. To this end, multi-layer heterostructure materials are produced by stacking 2D-crystals of each nanomaterial,<sup>20</sup> bonded by relatively weak van der Waals (vdW)-like forces. For example, the possibility of making a graphene/MoS<sub>2</sub> bilayer heterostructure has been demonstrated experimentally. Experimental findings show that synthesized materials are more thermally stable with high adsorption capacity.<sup>8</sup> Moreover, the space within the interlayer separation of these 2D nanosheets has often provided new and intriguing confined environments for molecular interactions due to the nanoconfinement effect.<sup>21–24</sup> The confined space between layers of these nanosheets has provided lower loss and stronger localization of active sites. In addition, the tunability of this spacing can enhance the physico-chemical properties of these nanostructures. Examples of these properties include surface area, penetration depth, adsorption capacity and catalytic activity. Therefore, understanding the underlying principles of CO<sub>2</sub> adsorption mechanisms within this nanoconfinement will provide in-depth knowledge into the materials science involving stacking routes of 2D nanomaterials.

The motivation to carry out this present study stems from the impact of interlayer orientation on the electronic properties and binding energy of substrate-grown monolayers<sup>25,26</sup> and multi-layered structures of the same material.<sup>27</sup> Also, experiments have shown that combining MoS<sub>2</sub> and graphene provides tremendous active sites for molecular adsorption that could responsibly remove smoke particles, CO and other toxic volatiles.<sup>8</sup> Therefore, it is possible to attune the structural parameters of these sites to facilitate CO<sub>2</sub> adsorption. Moreover, experimental and theoretical studies have shown that this nano-space presents new and extraordinary behaviours for molecular intercalations<sup>21</sup> and catalytic activities.<sup>22–24</sup> In this work, the effect of interlayer

distance between 2D materials, graphene/graphene (GrapheneB) and MoS<sub>2</sub>/MoS<sub>2</sub> (MoS<sub>2</sub>B) bilayers as well as graphene/MoS<sub>2</sub> (GMOs<sub>2</sub>) and MoS<sub>2</sub>/graphene (MoS<sub>2</sub>G) hybrids, on CO<sub>2</sub> adsorption mechanisms from first principles was investigated, with the aim of finding the most stable conformation for CO<sub>2</sub> adsorptions. The orientations and binding energies of CO<sub>2</sub> molecule on these surfaces are determined as well.

## 2 Computational methods and models

All calculations are carried out using DFT<sup>28</sup> in conjunction with the projector-augmented wave (PAW)<sup>29,30</sup> method, as implemented in Vienna *ab initio* simulation package (VASP).<sup>31,32</sup> The exchange and correlation potentials are treated with the generalized gradient approximation (GGA) functional of Perdew, Burke and Ernzerhof (PBE).<sup>33</sup> Valence electrons for Mo and S are generated in 4p<sup>6</sup>5s<sup>1</sup>4d<sup>5</sup> and 3s<sup>2</sup>3p<sup>4</sup> respectively, while the valence electron for C is generated in 2s<sup>2</sup>2p<sup>2</sup>. Different dispersion schemes are employed to understand the effects of vdW forces on the adsorption of CO<sub>2</sub>. Firstly, we include Grimme's<sup>34</sup> DFT-D2 method, which adds a semiempirical pairwise force field to conventional DFT calculations. Then, we employ the vdW-DF functional of Langreth and Lundqvist *et al.*, which adds a non-local correlation functional that approximately accounts for dispersion interaction.<sup>35–37</sup> The vdW-DFT methods with exchange-correlation energy given by revPBE,<sup>35</sup> the opt functionals (optPBE and optB88)<sup>38</sup> for accurate exchange functionals for vdW correlation, and vdW-DF2 of Langreth and Lundqvist groups<sup>39,40</sup> are considered.

Calculations are performed with a 4 × 4 × 1 supercell of monolayer MoS<sub>2</sub> (1H-MoS<sub>2</sub>) containing 16 Mo atoms and 32 S atoms, and a 5 × 5 × 1 supercell of monolayer graphene containing 50 C atoms for the monolayers. The MoS<sub>2</sub> bilayer contains two 4 × 4 × 1 supercells of monolayer MoS<sub>2</sub> (32 Mo atoms and 64 S atoms) while the graphene bilayer contains two 5 × 5 × 1 supercell monolayers of graphene (100 C atoms). For the hybrid structure, a supercell containing 5 × 5 × 1 lateral periodicity of graphene and 4 × 4 × 1 lateral periodicity of MoS<sub>2</sub> monolayer (16 Mo atoms, 32 S atoms and 50 C atoms) were employed which includes 1.9% lattice mismatch<sup>41</sup> in order to impose a commensurability condition between the graphene and MoS<sub>2</sub> monolayer. To illustrate the fundamental properties of the graphene–MoS<sub>2</sub> hybrid, we chose a lateral lattice parameter for the triangular lattice  $a = 12.34 \text{ \AA}$  that was optimized for isolated graphene, which is in agreement with graphene-based hybrid systems investigated previously<sup>41</sup> to ensure a smaller lattice mismatch. A large vacuum layer of 15 Å is used in the direction to the interface as an isolated slab boundary condition to avoid interlayer interactions. The test for cutoff energy and  $k$ -point grid is required for convergence and the result is summarized in the ESI.† The Gaussian smearing width was set to 0.2 eV. The total energy was converged to <10 meV for a plane-wave cutoff of 500 eV and 5 × 5 × 1 Monkhorst–Pack (MP)<sup>42</sup>  $k$ -point sampling for the Brillouin zone. Because of the large quasiparticle dynamics of graphene, the



Brillouin zone sampling of electronic states was performed using a finer MP mesh with size  $40 \times 40 \times 1$  per unit cell.<sup>43</sup> The tetrahedron method with Blöchl corrections for a pure graphene monolayer and bilayer DOS was also adopted. Finer MP mesh and Fermi smearing width of 0 eV was used to ensure accurate prediction of the DOS of semiconductors like graphene. For geometry relaxation, we used the method of conjugate gradient energy minimization. The convergence criterion for energy is chosen to be  $10^{-4}$  eV between two consecutive steps, and the maximum Hellmann–Feynman force exerting on each atom is less than  $0.03 \text{ eV } \text{Å}^{-1}$  upon ionic relaxation. Using Bader charge analysis, charge transfer between the substrate and the adsorbate is obtained.<sup>44</sup>

The adsorption energy of a  $\text{CO}_2$  molecule on multilayer and monolayer structures is investigated. Adsorption energy is calculated as:

$$E_{\text{AE}} = E_{\text{surf}+\text{CO}_2} - (E_{\text{surf}} + E_{\text{CO}_2}) \quad (1)$$

where  $E_{\text{surf}+\text{CO}_2}$  is the total energy of the optimized structure with an adsorbed  $\text{CO}_2$  molecule and  $E_{\text{surf}}$  and  $E_{\text{CO}_2}$  are the total energies of the pristine structure and isolated  $\text{CO}_2$  molecule respectively. A negative value of  $E_{\text{AE}}$  indicates that the adsorption is exothermic which means that the adsorption of  $\text{CO}_2$  molecules on the surface of the structures is favoured energetically.

## 3 Results and discussion

### 3.1 Bulk structure parameters

Graphene and  $\text{MoS}_2$  have hexagonal structures consisting of C–C and S–Mo–S layers respectively. Bulk graphene has two layers and each layer known as a unit cell consists of a central carbon atom connected to three carbon atoms with an adjacent carbon atom on both sides of the layers. Bulk  $\text{MoS}_2$  also has two such layers, with molybdenum (Mo) atoms of one layer directly above the sulphur (S) atoms of the other layer and *vice versa*. The lattice constant of pristine graphene and  $\text{MoS}_2$  sheets using a unit cell were calculated and compared with the literature in order to validate this method and its pseudopotentials. Calculated values are  $2.468 \text{ Å}$  and  $3.186 \text{ Å}$  for graphene and  $\text{MoS}_2$  respectively, which are in excellent agreement with previous theoretical results of  $2.463 \text{ Å}$  and  $3.20 \text{ Å}$ .<sup>45</sup> Based on this agreement, all other calculations were carried out, which are discussed in the subsequent sections.

### 3.2 Effects of interlayer distance on $\text{CO}_2$ adsorption

To obtain a clear understanding of the effect of the multilayer structures on  $\text{CO}_2$  adsorption, the adsorption energies of  $\text{CO}_2$  ( $E_{\text{AE}}$ ) at various interlayer distances of multilayer structures comprising graphene/graphene bilayer (GrapheneB),  $\text{MoS}_2/\text{MoS}_2$  bilayer ( $\text{MoS}_2\text{B}$ ), graphene/ $\text{MoS}_2$  hybrid ( $\text{GMoS}_2$ ) and  $\text{MoS}_2/\text{graphene}$  ( $\text{MoS}_2\text{G}$ ) hybrid were computed and the results are reported in Table 1. At the start of each geometry optimisation procedure,  $\text{CO}_2$  molecule is placed horizontally along the z-axis, at the middle distance between the two surface layers. This is assumed in order to avoid the repulsive effect between

Table 1 Adsorption energies of  $\text{CO}_2$  per carbon atom on different structures determined from different interlayer distances using the PBE of GGA<sup>43</sup>

Interlayer distance (Å)	Adsorption energies $E_{\text{AE}}$ (meV)			
	$\text{GMoS}_2$	$\text{MoS}_2\text{G}$	$\text{MoS}_2\text{B}$	GrapheneB
3.5	237.20	27.98	(NA)	179.98
6.0	61.10	44.87	(NA)	67.54
7.0	−17.10	−9.01	−36.00	−10.50
7.5	−23.55	−24.00	−38.00	−32.00
8.5	−24.11	−25.09	−28.25	−17.15
9.5	−15.76	−15.45	−16.90	−12.99
10.5	−8.90	−9.00	−8.13	−4.57
11.5	−9.76	−6.28	−4.76	−1.03
12.5	−2.54	−1.84	−2.30	0.68
13.5	−0.49	−0.55	−1.36	1.17

<sup>a</sup> NA: not adsorbed.

the electrons of the surface atoms and the lower oxygen atom of the  $\text{CO}_2$  molecule. Such a repulsive effect was commonly encountered in a configuration where the  $\text{CO}_2$  molecule is perpendicularly oriented to the surface. This was one of the initial configurations proposed for  $\text{CO}_2$  physisorption on pristine graphene<sup>46</sup> and graphene sheets with (0001) defects.<sup>47</sup>

As shown in Table 1, a more negative value means stronger  $\text{CO}_2$  adsorption. The positive values at separations below  $6.0 \text{ Å}$  indicate that the spacing between layers is so small that overlapping with  $\text{CO}_2$  occurred. Hence, energy is required to promote  $\text{CO}_2$  adsorption. According to Ataca *et al.*,<sup>45</sup> an interlayer spacing of  $6.145 \text{ Å}$  was estimated *via* DFT to be the suitable spacing for molecular interaction in the  $\text{MoS}_2$  interlayer. Below this, no adsorption (NA) will be observed, which corresponds to the results for  $\text{MoS}_2\text{B}$  in Table 1. Similarly, the most-suitable interlayer distance that permits molecular adsorption in GrapheneB has been determined to be between  $7 \text{ Å}$  and  $10 \text{ Å}$ .<sup>48</sup> However, unlike  $\text{MoS}_2\text{B}$ , energy is required to facilitate adsorption below this interlayer distance. This is attributed to the structural flexibility of graphene, in contrast to  $\text{MoS}_2$ . Furthermore, it can be stipulated that the required interlayer distance for  $\text{CO}_2$  adsorption in the  $\text{GMoS}_2$  hybrid should be the sum of the equilibrium distance between the  $\text{GMoS}_2$  substrate and the kinetic diameter of  $\text{CO}_2$  ( $3.30 \text{ Å}$ ).<sup>49</sup> The equilibrium distance between the  $\text{GMoS}_2$  substrate in different arrangements was measured to be around  $3.32 \text{ Å}$ .<sup>41,50</sup> Hence, energy is demanded for  $\text{CO}_2$  adsorption at interlayer distances less than  $6.62 \text{ Å}$  for the  $\text{GMoS}_2$  hybrid. Considering that  $\text{CO}_2$  molecule has a kinetic diameter of  $3.30 \text{ Å}$ ,<sup>49</sup> it is expected that  $\text{CO}_2$  intercalation should enlarge the distance between graphene sheets and  $\text{MoS}_2$  surface and that of the bilayer surfaces. By expanding the vdW gap between the two sheets, a charge transfer between the guest and the host surface is expected to occur with a very large repulsive energy.<sup>51</sup>

Fig. 1 demonstrates the variation of interlayer spacing with  $\text{CO}_2$  adsorption energies. This figure shows that  $7.5 \text{ Å}$  and  $8.5 \text{ Å}$  interlayer spacings are the most stable conformations for  $\text{CO}_2$  adsorption on the bilayer and hybrid structures respectively. Also, as the interlayer distance increases, the  $\text{CO}_2$



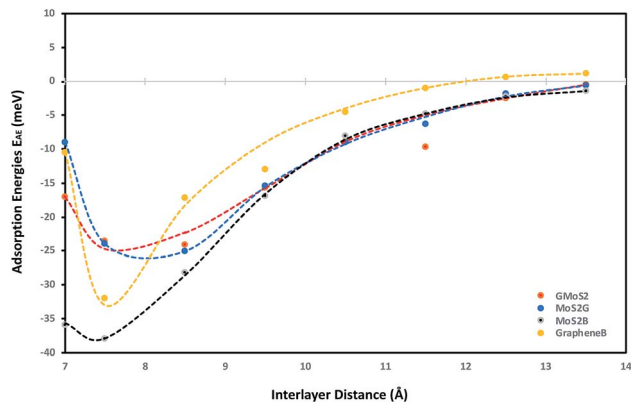


Fig. 1  $\text{CO}_2$  adsorption energy as a function of interlayer distance. Adsorption energy as a function of interlayer distance between the two monolayers for all the studied multilayer structures. Separate plots of each case are included in the ESI.

binding energy increases as well, but comes to a local minimum at 7.5 Å and 8.5 Å for the bilayer and hybrid structures respectively. It is estimated that GrapheneB has the least total adsorption capacity given the considered range of interlayer distance. It can be assumed that the presence of  $\text{MoS}_2$  in the hybrid buttressed  $\text{CO}_2$  adsorption beyond interlayer spacing greater than 8.0 Å. The plot for the  $\text{GMoS}_2$  hybrid structure is in good agreement with the work of Ma *et al.*,<sup>41</sup> although only graphene adhesion on  $\text{MoS}_2$  was considered. The peak binding energies per C atom for  $\text{GMoS}_2$ ,  $\text{MoS}_2\text{G}$ ,  $\text{MoS}_2\text{B}$  and GrapheneB are  $-24.11$  meV,  $-38.00$  meV,  $-25.09$  meV and  $-32.00$  meV, respectively. Moreover, the adsorption energies of these multilayer structures seem to be a simple sum of the contributions from the constituents with decreasing binding energies in the order:  $\text{MoS}_2\text{B} > \text{GrapheneB} > \text{MoS}_2\text{G} > \text{GMoS}_2$ . These values are low which correspond to weak interactions between the layers and  $\text{CO}_2$  molecule. This is due to the inability of conventional DFT approaches to describe dispersion force in adsorption systems. These values were improved when factors like  $\text{CO}_2$  orientation and addition of vdW interactions between the  $\text{CO}_2$  molecule and the surfaces were considered. Furthermore, two different configurations of the bilayer heterostructure were considered. This is to explain the effect of structural anisotropy on the adsorption energies of heterostructures. In the first structure graphene is the substrate ( $\text{GMoS}_2$ ) while in the second case  $\text{MoS}_2$  is the substrate ( $\text{MoS}_2\text{G}$ ). Optimization results depict that there is discrepancy in the binding energies of  $\text{CO}_2$  adsorption on these hybrid structures ( $\text{GMoS}_2$  and  $\text{MoS}_2\text{G}$ ). This can be ascribed to the modality of the atom arrangement (C–Mo–S–O and Mo–S–O–C) just like in the case of zigzag and armchair arrangements of bilayer graphene<sup>27</sup> and the interlayer orientation effect on bilayer heterostructures. This indicates that the conformational anisotropy of this structure determines to a certain degree its  $\text{CO}_2$  adsorption energy. The most appropriate orientation of this hybrid for optimal adsorption performance is subject to further investigation.

### 3.3 Effects of $\text{CO}_2$ orientation and position

To further understand the interaction of  $\text{CO}_2$  within these multilayer structures, we calculated  $\text{CO}_2$  adsorption energy at different  $\text{CO}_2$  orientations and positions using the same optimum interlayer distance presented earlier. Table 2 summarizes the calculated adsorption energies per carbon atom for the most stable adsorption configuration of the multilayer structures in comparison with the most stable monolayers. In all calculations,  $\text{CO}_2$  was placed at the mid-point of the interlayer distance for the multilayer structure and the same value was chosen for the monolayers for proper comparison. Also, since the test for cutoff energy and  $k$ -point grid is required for convergence, we calculated total energies using different cutoffs and  $k$ -points and the results are summarized in the ESI.<sup>†</sup> The total energy was converged to  $<10$  meV for a plane wave cutoff of 500 eV and  $5 \times 5 \times 1$  Monkhorst–Pack<sup>42</sup>  $k$ -point sampling for the Brillouin zone. The adsorption energies of the multilayer structures also seem to be a simple sum of the contributions from the constituents. The structures displayed low energies, which explains the difficulty of conventional DFT approaches to accurately reflect the vdW force in adsorption systems. Our goal here is to gain insight into the surface most active site, which is the  $\text{CO}_2$  adsorption energies with respect to its positions and orientation on the surface. From the obtained results, it is evident that  $\text{CO}_2$  molecule prefers to interact with a parallel or inclined orientation with the adsorbent surface than the perpendicular orientation. For instance, considering the edge position of GrapheneB bilayer, the adsorption energy for parallel and  $30^\circ$  parallel-rotation are  $-27.05$  and  $-23.62$  meV respectively, but the value is  $-5.65$  meV for perpendicular attack of  $\text{CO}_2$  at the same position. This is due to the repulsive effect between the electrons in C-atoms (graphene) and the lower O-atom ( $\text{CO}_2$  molecule). Consequently, the molecule stays less bounded to the surface compared to the optimum adsorption site. The same effect was addressed by Cabrera-Sanfeliix, when  $\text{CO}_2$  was placed initially perpendicular on graphene sheets with defects.<sup>47</sup> Parallel attack of  $\text{CO}_2$  at the edge position is observed to be the most favourite adsorption site for  $\text{GMoS}_2$ ,  $\text{MoS}_2\text{B}$  and  $\text{MoS}_2\text{G}$  composites with stable adsorption energies of  $-27.16$ ,  $-40.37$  and  $-27.07$  meV respectively, while the centre position was the most favourable adsorption site for GrapheneB ( $-32.37$  meV).  $\text{GMoS}_2$  and  $\text{MoS}_2\text{G}$  retained this position as their active site for  $\text{CO}_2$  inclined ( $30^\circ$  and  $45^\circ$  respectively) attack on the surface, in contrast to GrapheneB, which favoured the centre position as the favourite adsorption site. These differences can be attributed to changes of the electronic environment causing charge redistribution culminating to variation in adsorption capabilities. In addition, the adsorption capacity derived from the edge site of  $\text{MoS}_2$  (ref. 52 and 53) can contribute to these changes. For instance,  $\text{MoS}_2$  monolayer ( $\text{M}4 \times 4$ ) favoured the edge position as the favourite adsorption site while graphene monolayer ( $\text{G}5 \times 5$ ) favoured the centre position as the favourite adsorption site. Therefore, it is expected that the edge position will be the most favourite adsorption site for the  $\text{GMoS}_2$  and  $\text{MoS}_2\text{G}$  due to the inclusion of the  $\text{MoS}_2$  layer, whose catalytic activity occurs at the edge





Table 2 Adsorption energies of possible initial configurations of CO<sub>2</sub> adsorption on perfect multilayer and monolayer structures using PBE<sup>a,33</sup>

Position/orientation to the surface	Adsorption energies $E_{AE}$ (meV)					
	GMoS <sub>2</sub>	MoS <sub>2</sub> G	MoS <sub>2</sub> B	GrapheneB	G5 × 5	M4 × 4
Centre/parallel	-25.49	-26.98	-27.99	-26.47	-15.15	-17.13
Edge/parallel	-26.48	-26.54	-40.37	-27.05	-13.21	-20.33
Centre/perpendicular	11.27	11.77	100.02	-25.64	-3.10	-9.92
Edge/perpendicular	38.00	47.97	-27.31	-5.65	-3.90	-4.23
Centre/parallel (rotated 30°)	-23.27	-26.73	-37.44	-26.04	-16.38	-16.60
Edge/parallel (rotated 30°)	-27.16	-26.60	-35.98	-23.62	-13.94	-19.37
Centre/parallel (rotated 45°)	-24.03	-25.05	-37.75	-32.37	-14.07	-17.47
Edge/parallel (rotated 45°)	-26.16	-27.07	-35.98	-28.23	-14.41	-19.23

<sup>a</sup> G5 × 5 and M4 × 4 are 5 × 5 and 4 × 4 supercells of graphene and MoS<sub>2</sub> monolayers respectively.

sites. Also, since M4 × 4 has higher adsorption energy (-20.33 meV) than G5 × 5 (-16.38 meV), it is expected that M4 × 4 will determine to a certain extent the adsorption properties of the hybrid structures (GMoS<sub>2</sub> and MoS<sub>2</sub>G).

Improved adsorption energies were obtained when the dispersion correction to DFT is considered. To take dispersion force into account, we carried out a spin-polarized calculation using different dispersion force methods and compared the results with conventional DFT approaches. The adsorption energies significantly improved by incorporating vdW interactions between the CO<sub>2</sub> molecule and the surfaces as shown in Table 3. However, there are no detailed experimental or computational data for CO<sub>2</sub> adsorption on hybrid and bilayer structures so far. Based on previous calculations of CO<sub>2</sub> adsorption on pristine monolayer graphene<sup>46,54</sup> and MoS<sub>2</sub>,<sup>55</sup> the results obtained using optPBE functional for the monolayers should be reliable. Also, we find that CO<sub>2</sub> binds strongly on these multilayer structures. In fact, graphene bilayer gives the highest CO<sub>2</sub> adsorption energy among the multilayer structures when vdW-DF2 interactions are included, in contrast to conventional PBE results. This is in contrast to the previous suggestion from conventional PBE results that the adsorption energy of the bilayers is approximately a simple sum of the contributions from the constituents. In addition, it is expected that adsorption energy of CO<sub>2</sub> on perfect graphene and MoS<sub>2</sub> sheets should be lower because of lack of dispersive interactions

in DFT. Due to the addition of vdW interactions, there is a tendency for overestimation of molecular adsorption.<sup>56,57</sup> For instance, vdW-DF2 correlation overestimates the adsorption energy of graphene bilayer, while revPBE and optPBE correlations overestimate the adsorption energy of MoS<sub>2</sub> bilayer. Furthermore, it is observed that CO<sub>2</sub> adsorption energies obtained when revPBE and optPBE correlations are included are similar. Likewise, there are similarities in the adsorption energies calculated using DFT-D2 and vdW-DF2 correlations except for the adsorption energy of bilayer graphene which is overestimated by vdW-DF2 correlation. Also, the results of these correlations for the MoS<sub>2</sub> monolayer are in good agreement with the previous calculation.<sup>55</sup> On the other hand, the stronger binding between CO<sub>2</sub> and graphene monolayer when using DFT-D2 correlation is in agreement with reported values given for CO<sub>2</sub> adsorption on perfect monolayer graphene, ~147–151 meV.<sup>56,57</sup> Generally, the increase in CO<sub>2</sub> adsorption energy with the introduction of vdW interactions indicates that vdW interaction dominates during the adsorption process. This is comparable with previous theoretical results that incorporated vdW interaction.<sup>1,55</sup>

### 3.4 Structural parameters

The optimized adsorption configuration of CO<sub>2</sub> on most stable multi-layered structures determined by PBE calculations and the stacking type of the structures are shown in Fig. 2. The bond length (C=O), angle of CO<sub>2</sub> molecule, interlayer distances, C=

Table 3 Comparison of the adsorption energies ( $E_{AE}$ , meV) of CO<sub>2</sub> on bilayer, hybrid and monolayer structures determined from different methods

Structures	This work						Reference <sup>a</sup>	
	PBE	DFT-D2	revPBE	optPBE	optB88	vdW-DF2	PBE	DFT-D2
GMoS <sub>2</sub>	-27.16	-145.17	-321.73	-333.01	-242.13	-161.31	—	—
MoS <sub>2</sub> G	-27.07	-169.14	-300.47	-313.54	-254.74	-186.46	—	—
MoS <sub>2</sub> B	-40.37	-195.33	-1157.28	-1069.94	-829.09	-232.65	—	—
GrapheneB	-32.37	-241.34	-405.23	-450.26	-368.47	-1694.0	—	—
G5 × 5	-16.38	-153.30	-209.58	-249.98	-224.41	-160.21	-17.10(ref. 46)	-60(ref. 1)
M4 × 4	-20.33	-126.30	-201.25 (-210 (ref. 55))	-232.40 (-253 (ref. 55))	-204.50	-152.38	-4.00 (ref. 55)	-139.0 (ref. 55)

<sup>a</sup> The corresponding values of previous calculations for monolayer graphene and MoS<sub>2</sub> are given for the sake of comparison. The PBE adsorption energy of ref. 55 for MoS<sub>2</sub> monolayer appears to be too small, this is because ref. 55 did not consider CO<sub>2</sub> orientation, a similar value was obtained when we considered CO<sub>2</sub> orientation (Table 2). Similarly, the DFT-D2 adsorption energy of ref. 1 for graphene monolayer is too small due to the same reason. But the revPBE and optPBE adsorption energies of MoS<sub>2</sub> monolayer are in good agreement with the adsorption energy of ref. 55.



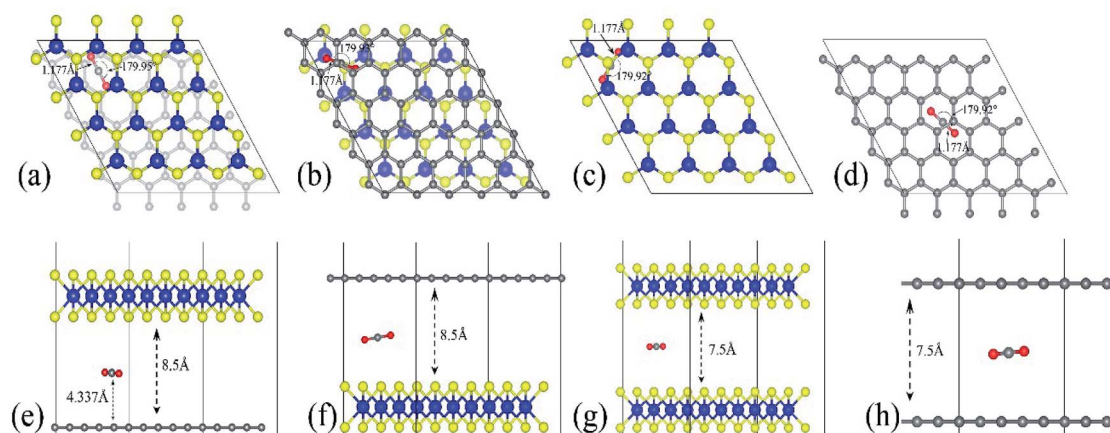


Fig. 2 The most stable configuration of the multilayered structures with adsorbed  $\text{CO}_2$ . (a–d) are the top views of  $\text{GMoS}_2$ ,  $\text{MoS}_2\text{G}$ ,  $\text{MoS}_2\text{B}$  and GrapheneB and (e and f) are side views of the same structures respectively. Color code: Mo, blue, S, yellow, C, grey, and O, red. The lines define the interlayer distances, bond lengths, and molecular distance while the curved lines define the bond angle. AA stacking type is chosen for both  $\text{MoS}_2\text{B}$  and GrapheneB. This is because AA stacking of graphene has been shown to be more preferable for intercalation of molecules according to ref. 59 and ref. 60. Although the binding energy of AB stacking of graphene bilayer is lower than that of AA stacking, which makes AB stacking more stable than AA stacking, AA stacking intercalation structures are more favorable than AB stacking ones, ref. 60. For the hybrid structures, the stacking type is chosen according to ref. 50, where one C atom in the unit cell of graphene sits exactly below a Mo atom. It is reported in ref. 41 that another configuration called TS, where the stacking is such that C atom sits below an S atom, is equivalent in both the binding and electronic properties.

C, Mo=S bond lengths and the molecular distance are presented in this figure. To further understand the interaction between  $\text{CO}_2$  and the surface, several structural parameters are calculated and compared with previous studies. The calculated bond length and angle for free  $\text{CO}_2$  molecule are 1.177 Å and  $179.87^\circ$  respectively, which is in line with 1.16 Å and  $180^\circ$  from experiments,<sup>58</sup> and 1.175 Å and  $178^\circ$  from previous theoretical calculation.<sup>46</sup> Based on experimental values, bond angles of adsorbed  $\text{CO}_2$  decreased by about 0.05–0.13°, depending on the structures of the nanocomposite (Table 4). There is no chemical bonding observed, and interactions are dominated by weak vdW forces. In other words, the entire adsorption is purely physisorption due to the small adsorption energy and large separation height (molecular distance). Moreover, only the stable configurations obtained from PBE calculations are

presented. This is because the inclusion of vdW interactions slightly changes the stable configurations.

Bader charge analysis shows that the interaction between  $\text{CO}_2$  and these surfaces causes charge transfer and redistributions (Fig. 3). The amount of charge transferred from Bader charge analysis is summarized in Table 4. It can be seen that while 0.028e of charge is transferred to  $\text{CO}_2$  on the  $\text{GMoS}_2$  surface where graphene is the bottommost layer, 0.810e of charge is transferred from  $\text{CO}_2$  to  $\text{MoS}_2$  where  $\text{MoS}_2$  is the topmost layer. This could indicate that the modality of the atom arrangement (C–Mo–S–O and Mo–S–O–C) in both structures affects the charge transfer between the molecule and surface atoms. In other words, the conformational anisotropy of this structure determines to a certain point the charge transfer and redistribution just like in its  $\text{CO}_2$  adsorption energy. Meanwhile, in some cases, the adjacent S atom of  $\text{MoS}_2$  and C atom of

Table 4 Amount of charge transferred to  $\text{CO}_2$  molecule (single atom and  $\text{CO}_2$  molecule), amount of charge transferred from adjacent Mo, S and C atoms on the surface nearer to the adsorbed  $\text{CO}_2$  molecule, and the change in bond length and bond angle after  $\text{CO}_2$  adsorption (length and angle, with angle change in parentheses)<sup>a</sup>

Structures	Amount of charge transferred (e)							Bond length (Å) angle (deg)	
	O1	C	O2	$\text{CO}_2$	Mo	S	Cg	C=O	O=C=O
$\text{GMoS}_2$	−0.001	−0.018	−0.009	−0.028	0.001	−0.002	−0.001	1.18	179.95 (0.05)
$\text{MoS}_2\text{G}$	0.804	0.036	−0.030	0.810	0.000	0.000	−0.003	1.18	179.93 (0.07)
$\text{MoS}_2\text{B}$	0.035	−0.033	0.015	0.017	0.000	0.000	N/A	1.18	179.92 (0.08)
GrapheneB	0.021	−0.012	−0.016	−0.007	N/A	N/A	−0.004	1.18	179.92 (0.08)
$\text{G5} \times 5$	0.001	0.020	−0.015	0.006	N/A	N/A	−0.004	1.18	179.89 (0.11)
$\text{M4} \times 4$	0.035	−0.047	0.002	−0.010	0.000	0.084	N/A	1.18	179.87 (0.13)

<sup>a</sup> Bader charge analysis is used for the charge transfer calculations. Cg is the adjacent carbon atom of graphene surface that is nearer to the adsorbed  $\text{CO}_2$  molecule, while Mo and S are adjacent molybdenum and sulphur atoms of the  $\text{MoS}_2$  surface that are nearer to the adsorbed  $\text{CO}_2$  molecule. The adjacent atoms were chosen arbitrarily based on their proximity to the adsorbed  $\text{CO}_2$  molecule. Negative charge denotes electron gain.



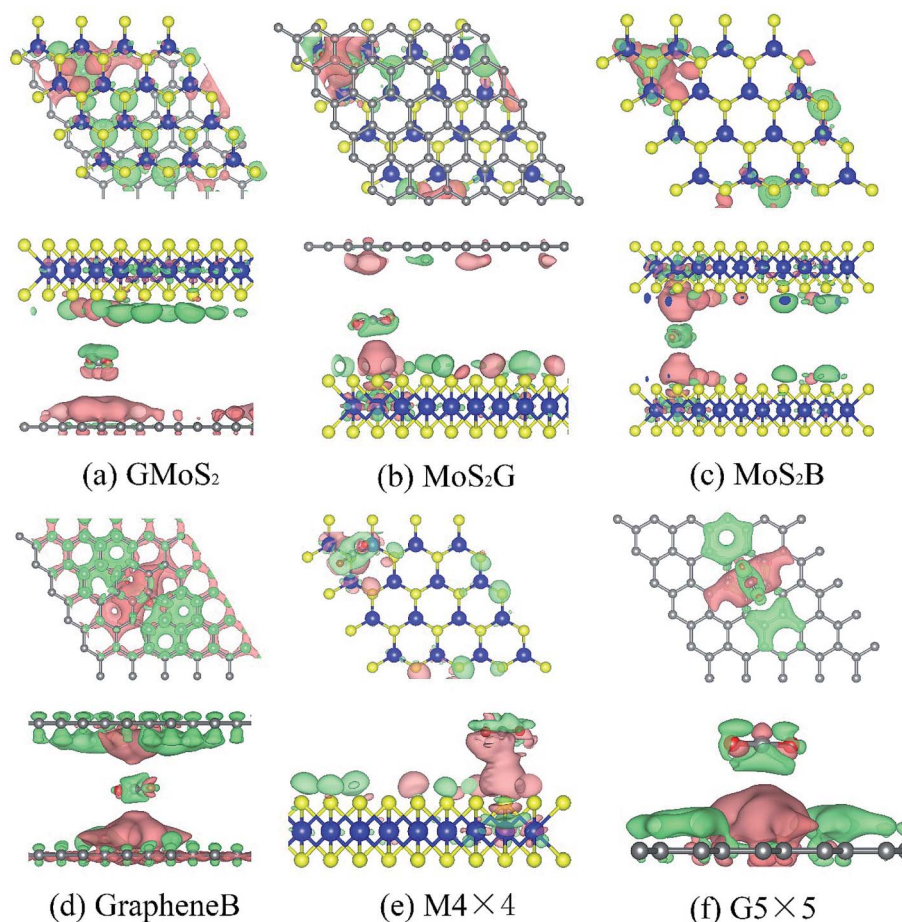


Fig. 3 Top and side views of charge density difference plots for (a) graphene/MoS<sub>2</sub> hybrid, (b) MoS<sub>2</sub>/graphene hybrid, (c) MoS<sub>2</sub> bilayer, (d) graphene bilayer, (e) 4 × 4 supercell MoS<sub>2</sub> monolayer, and (f) 5 × 5 supercell graphene monolayer. The pink and green distributions correspond to charge accumulation (electron excess) and depletion (electron loss) respectively. Isosurfaces:  $\pm 5 \times 10^{-5} e^{-1} \text{ \AA}^3$ .

graphene nearer to the adsorbed CO<sub>2</sub> gain electrons, while Mo atom loses an electron in all cases. This is because of the higher electronegativity of S and C atoms compared to the Mo atom. Fig. 3 presents the charge density difference plots for CO<sub>2</sub> molecule-surfaces obtained using the formula:

$$\Delta\rho = \rho_{\text{mole+surf}} - (\rho_{\text{mole}} + \rho_{\text{surf}}) \quad (2)$$

where  $\rho_{\text{mole+surf}}$ ,  $\rho_{\text{mole}}$  and  $\rho_{\text{surf}}$  are the charge densities of non-interacting components (CO<sub>2</sub> and surfaces), isolated CO<sub>2</sub> molecule and pure surface of the structures, respectively.

The pink region shows the charge accumulation (electron excess) while the green region indicates the charge depletion (electron loss). In all the structures, it can be shown that there is a noticeable polarization of the surfaces upon CO<sub>2</sub> adsorption, and electrostatic interaction plays a role in the attractive interaction. Moreover, there is charge accumulation on GMoS<sub>2</sub>, MoS<sub>2</sub>G, MoS<sub>2</sub>B, GrapheneB, G5 × 5 and M4 × 4 upon CO<sub>2</sub> adsorption, which suggests that CO<sub>2</sub> also has a charge-donor characteristic. In addition, polarization in GrapheneB and MoS<sub>2</sub>B structures showed to be stronger than in MoS<sub>2</sub>G, G5 × 5, GMoS<sub>2</sub> and MoS<sub>2</sub> structures, which gives rise to larger adsorption energies. This explains why GrapheneB and MoS<sub>2</sub>B

displayed larger adsorption energies (−241 and −195 meV, respectively) than MoS<sub>2</sub>G, G5 × 5, GMoS<sub>2</sub> and MoS<sub>2</sub> (−169, −153, −145, and −126 meV, respectively) after spin-polarized DFT-D2 calculations mentioned earlier.

### 3.5 Density of states (DOS) plots

To analyse the electronic properties of the adsorbed CO<sub>2</sub> interaction on the multilayer structures, the total electronic density of states (DOS) was studied for graphene and MoS<sub>2</sub> monolayer and bilayer surfaces before and after CO<sub>2</sub> adsorption. Fig. 4 shows that the DOS spectra for either the valence or conduction band of the monolayer and bilayer surfaces do not show significant changes before and after adsorption. This indicates that the CO<sub>2</sub> interaction with these surfaces is purely physisorption, involving weak interaction forces such as vdW and Lewis acid–base interaction. Subsequently, this will not affect the electronic structure of surface atoms significantly. Although it is observed that the adsorption of CO<sub>2</sub> introduces several distinct states at the lower-lying valence bands within the energy level of −22 eV to −25 eV, and offsets some peaks at the conduction bands within energy levels 2.5–5.0 eV, there is no noticeable modification of the DOS near the Fermi level,





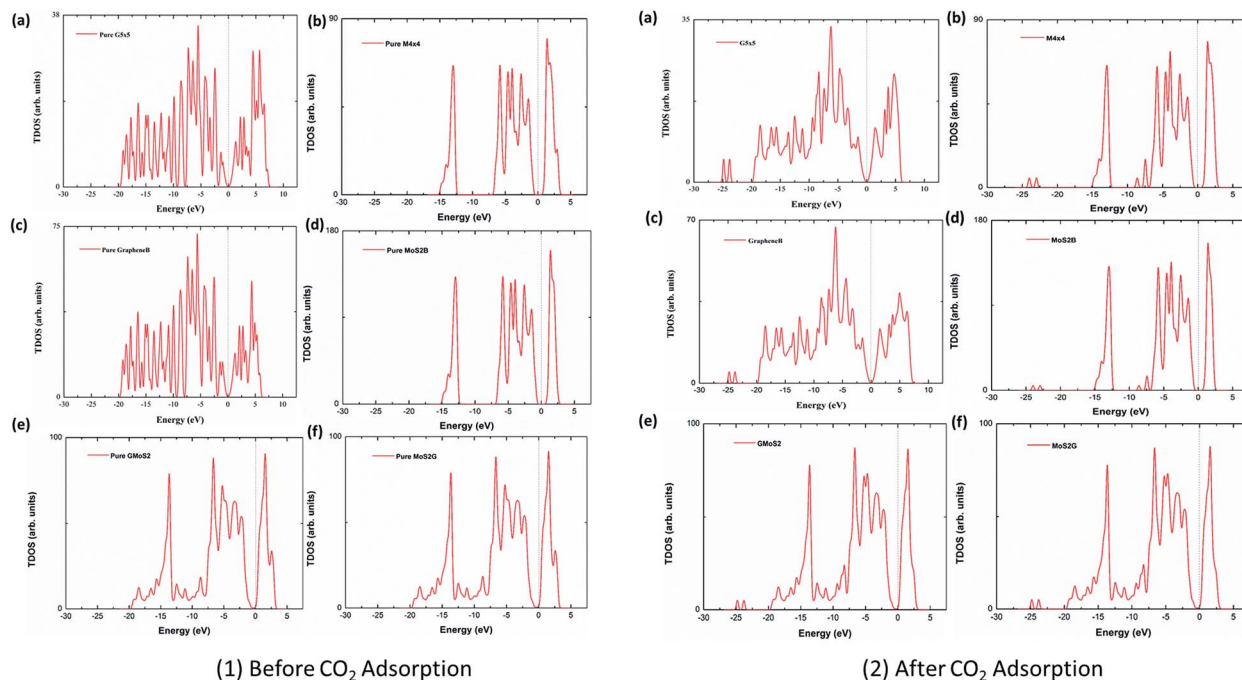


Fig. 4 Total DOS spectra before (1) and after (2) CO<sub>2</sub> adsorption. (a) 5 × 5 supercell graphene monolayer, (b) 4 × 4 supercell MoS<sub>2</sub> monolayer, (c) graphene bilayer, (d) MoS<sub>2</sub> bilayer, (e) graphene/MoS<sub>2</sub> hybrid, and (f) MoS<sub>2</sub>/graphene hybrid. The dashed line represents the Fermi level.

which also concludes that CO<sub>2</sub> physisorption does not have a substantial effect on the electronic properties of graphene and MoS<sub>2</sub>. DOS spectra of graphene (G5 × 5) and MoS<sub>2</sub> (M4 × 4) monolayer, bilayers (GrapheneB and MoS<sub>2</sub>B) and hybrid (GMoS<sub>2</sub> and MoS<sub>2</sub>G) structures were compared. It is observed that the DOS spectra of the monolayer and bilayer structures are nearly the same. MoS<sub>2</sub> (monolayer and bilayer) displayed a spectrum of an intrinsic semiconductor with direct bandgap of 1.66 eV before and after CO<sub>2</sub> adsorption, which fairly correspond to both previous experimental (1.80 eV) and computational (1.58 eV) results obtained for pure MoS<sub>2</sub> monolayer, whereas graphene (monolayer and bilayer) exhibited a semi-metal spectrum with a small overlap between the valence and conduction band (zero bandgap material).<sup>61,62</sup> However, it is noted here that 5 × 5 × 1 *k*-point mesh produced an un-converged sampling of Brillouin zone for pure graphene, and introduced small peaks around the Fermi level. This is attributed to the large quasiparticle dynamics of graphene,<sup>43</sup> particularly at fewer *k*-point meshes like 5 × 5 × 1. To correct this anomaly, a very fine *k*-point mesh without increasing the computational cost was proposed. Accordingly, a finer MP mesh of size 40 × 40 × 1 in the unit cell and the tetrahedron method with Blöchl corrections was used to reproduce the Brillouin zone sampling of electronic states. The 40 × 40 × 1 *k*-point mesh centered at  $\Gamma$  gave a faster convergence and accurately predicted the DOS of pure graphene. It also eliminated the small peaks around the Fermi level. As shown in Fig. 4, the Dirac cone was obtained in pure graphene monolayer and bilayer, which is unaffected after CO<sub>2</sub> adsorption. The plots are in good agreement with the work of ref. 63, which tested different *k*-point meshes to generate the DOS of graphene

monolayer and bilayer structures. However, the DOS spectra of the bilayer heterostructure seem to be a simple sum of those of each constituent with a shift of the valence and conduction bands. The conduction band shifted closer to the Fermi level, a characteristic of an n-type semiconductor. There is a noticeable modification of DOS spectra near the Fermi level, as the conduction band shifted closer to the Fermi level. This indicates a small band gap opening due to the variation of on-site energy induced by MoS<sub>2</sub> that is found in graphene/MoS<sub>2</sub> bilayer heterostructures.<sup>41</sup> As shown in Fig. 4 (1 and 2), the calculated bandgap is 0.04 eV irrespective of the modality of the atom arrangement in both structures. This specifies that the graphene monolayer in the heterostructure loses its metallic nature, and massless electron characteristics and becomes more semiconducting with a direct narrow bandgap, while significantly increasing the conductivity of the MoS<sub>2</sub> monolayer. This is comparable with other hybrid structures.<sup>41</sup> It is also observed that the DOS spectra of the hybrid structures are the same, which indicates that the conformational anisotropy does not have any effect on the DOS spectra bilayer heterostructures unlike in its CO<sub>2</sub> adsorption energy.

## 4 Conclusion

In this work, a comprehensive first-principles study of the CO<sub>2</sub> adsorption mechanism on graphene/graphene, MoS<sub>2</sub>/MoS<sub>2</sub> bilayers and graphene/MoS<sub>2</sub>, MoS<sub>2</sub>/graphene hybrids using GGA-PBE is presented. CO<sub>2</sub> adsorption energies, indicative of adsorption capacity, were calculated at various interlayer spacings. 7.5 Å and 8.5 Å interlayer spacings are the most stable conformations for CO<sub>2</sub> adsorption on the bilayer and hybrid





structures respectively, with the bilayers displaying higher adsorption capacities. Beyond an interlayer spacing of 8.0 Å, CO<sub>2</sub> adsorption was promoted by the presence of MoS<sub>2</sub> in the hybrid. However, this varied with the configuration of the bilayer structure. Moreover, the most active adsorption site was determined by varying CO<sub>2</sub> molecule orientation. A parallel CO<sub>2</sub> attack at the edge position of the surface exhibited the highest adsorption energies. Parallely inclined orientations also showed promising adsorption potential, however this varied in attack positions for each nanocomposite. The total electronic density of states analysis reveals that CO<sub>2</sub> interaction with these surfaces is purely physisorption, which mainly involves weak interaction forces. The Bader charge analysis indicates that there is charge transfer and redistribution between the substrate and the adsorbate. In addition, it was demonstrated that conformational anisotropy could affect CO<sub>2</sub> adsorption and other properties due to the modality of the atom arrangement (C–Mo–S–O and Mo–S–O–C) in both heterostructures. In general, the modification of interlayer spacing and structural configuration during the synthesis route is the key to enhancing the adsorption performance of these nanocomposites. This is attainable given the improving successes in atomic level studies and combination of the design of experiments like nanoparticle tracking analysis. Finally, the incorporation of vdW interactions boosted the adsorption energies, thus portraying the importance of doping and functionalization in improving the adsorption performance of these composites.

## Conflicts of interest

There is no conflicts to declare.

## Acknowledgements

Following funding bodies are acknowledged for partially sponsoring this research: National Key Research and Development Projects (2017YFC0210400 and 2017YFB0603202), National Natural Science Foundation of China (51706114). Special thanks to the Advanced Energy and Environmental Materials Technology (AEEMT) research group at the University of Nottingham Ningbo China. The computational resources are fully provided by the High-Performance Computer (HPC) System of UNNC. The author (Francis Enujekwu) thanks Bamidele Akinwolemiwa for his helpful discussions.

## References

- 1 J. Li, *et al.*, Enhanced CO<sub>2</sub> capture on graphene *via* N, S dual-doping, *Appl. Surf. Sci.*, 2017, **399**, 420–425.
- 2 S. A. Tawfik, *et al.*, Multiple CO<sub>2</sub> capture in stable metal-doped graphene: a theoretical trend study, *RSC Adv.*, 2015, **5**(63), 50975–50982.
- 3 C. Cazorla, The role of density functional theory methods in the prediction of nanostructured gas-adsorbent materials, *Coord. Chem. Rev.*, 2015, **300**, 142–163.
- 4 NOAA, *Trends in Carbon Dioxide: Globally averaged marine surface monthly mean data*, Earth System Research

Laboratory (ESRL), [ftp://aftp.cmdl.noaa.gov/products/trends/co2/co2\\_mm\\_gl.txt](ftp://aftp.cmdl.noaa.gov/products/trends/co2/co2_mm_gl.txt), [www.esrl.noaa.gov/gmd/ccgg/trends/](http://www.esrl.noaa.gov/gmd/ccgg/trends/), 2017.

- 5 R. F. Service, The Carbon Conundrum, *Science*, 2004, **305**(5686), 962–963.
- 6 A. B. Rao and A. R. Rubin, A Technical, Economic, and Environmental Assessment of Amine-Based CO<sub>2</sub> Capture Technology for Power Plant Greenhouse Gas Control, *Environ. Sci. Technol.*, 2002, **36**(20), 4467–4475.
- 7 L. Ci, *et al.*, Atomic layers of hybridized boron nitride and graphene domains, *Nat. Mater.*, 2010, **9**(5), 430.
- 8 D. Wang, *et al.*, Space-Confined Growth of Defect-Rich Molybdenum Disulfide Nanosheets Within Graphene: Application in The Removal of Smoke Particles and Toxic Volatiles, *ACS Appl. Mater. Interfaces*, 2016, **8**(50), 34735–34743.
- 9 L. Ci, *et al.*, Atomic layers of hybridized boron nitride and graphene domains, *Nat. Mater.*, 2010, **9**, 430.
- 10 C. Lee, *et al.*, Frictional Characteristics of Atomically Thin Sheets, *Science*, 2010, **328**(5974), 76–80.
- 11 K. S. Novoselov, *et al.*, Two-dimensional atomic crystals, *Proc. Natl. Acad. Sci. U. S. A.*, 2005, **102**(30), 10451–10453.
- 12 B. K. Miremadi and S. R. Morrison, Highactivity catalyst from exfoliated MoS<sub>2</sub>, *J. Catal.*, 1987, **103**(2), 334–345.
- 13 C. Lee, *et al.*, Anomalous Lattice Vibrations of Single- and Few-Layer MoS<sub>2</sub>, *ACS Nano*, 2010, **4**(5), 2695–2700.
- 14 K. F. Mak, *et al.*, Atomically Thin MoS<sub>2</sub>: A New Direct-Gap Semiconductor, *Phys. Rev. Lett.*, 2010, **105**(13), 136805.
- 15 B. Radisavljevic, *et al.*, Single-layer MoS<sub>2</sub> transistors, *Nat. Nanotechnol.*, 2011, **6**, 147.
- 16 Y. Zhang, *et al.*, Ambipolar MoS<sub>2</sub> Thin Flake Transistors, *Nano Lett.*, 2012, **12**(3), 1136–1140.
- 17 L. Hai, *et al.*, Fabrication of Single- and Multilayer MoS<sub>2</sub> Film-Based Field-Effect Transistors for Sensing NO at Room Temperature, *Small*, 2012, **8**(1), 63–67.
- 18 Q. He, *et al.*, Fabrication of Flexible MoS<sub>2</sub> Thin-Film Transistor Arrays for Practical Gas-Sensing Applications, *Small*, 2012, **8**(19), 2994–2999.
- 19 H. Zhao, *et al.*, Structural defects in 2D MoS<sub>2</sub> nanosheets and their roles in the adsorption of airborne elemental mercury, *J. Hazard. Mater.*, 2019, **366**, 240–249.
- 20 A. K. Geim and I. V. Grigorieva, Van der Waals heterostructures, *Nature*, 2013, **499**, 419.
- 21 R. Mu, *et al.*, Visualizing chemical reactions confined under graphene, *Angew. Chem., Int. Ed.*, 2012, **51**(20), 4856–4859.
- 22 X. Pan and X. Bao, The Effects of Confinement inside Carbon Nanotubes on Catalysis, *Acc. Chem. Res.*, 2011, **44**(8), 553–562.
- 23 B. Smit and T. L. M. Maesen, Towards a molecular understanding of shape selectivity, *Nature*, 2008, **451**, 671.
- 24 J. M. Thomas and R. Raja, Exploiting Nanospace for Asymmetric Catalysis: Confinement of Immobilized, Single-Site Chiral Catalysts Enhances Enantioselectivity, *Acc. Chem. Res.*, 2008, **41**(6), 708–720.
- 25 B. B. Kappes, *et al.*, Orientation-dependent binding energy of graphene on palladium, *Appl. Phys. Lett.*, 2013, **102**(5), 051606.
- 26 Y. Murata, *et al.*, Orientation-dependent work function of graphene on Pd(111), *Appl. Phys. Lett.*, 2010, **97**(14), 143114.



- 27 Y. Murata, *et al.*, Growth structure and work function of bilayer graphene on Pd(111), *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, **85**(20), 205443.
- 28 P. Hohenberg and W. Kohn, Inhomogeneous Electron Gas, *Phys. Rev.*, 1964, **136**(3B), B864–B871.
- 29 P. E. Blöchl, Projector augmented-wave method, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1994, **50**(24), 17953–17979.
- 30 G. Kresse and D. Joubert, From ultrasoft pseudopotentials to the projector augmented-wave method, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1999, **59**(3), 1758–1775.
- 31 G. Kresse and J. Furthmüller, Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**(16), 11169–11186.
- 32 G. Kresse and J. Furthmüller, Efficiency of ab-initio total energy calculations for metals and semiconductors using a plane-wave basis set, *Comput. Mater. Sci.*, 1996, **6**(1), 15–50.
- 33 J. P. Perdew, K. Burke and M. Ernzerhof, Generalized Gradient Approximation Made Simple, *Phys. Rev. Lett.*, 1996, **77**(18), 3865–3868.
- 34 S. Grimme, Semiempirical GGA-type density functional constructed with a long-range dispersion correction, *J. Comput. Chem.*, 2006, **27**(15), 1787–1799.
- 35 M. Dion, *et al.*, Van der Waals Density Functional for General Geometries, *Phys. Rev. Lett.*, 2004, **92**(24), 246401.
- 36 J. Klimeš, D. R. Bowler and A. Michaelides, Van der Waals density functionals applied to solids, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **83**(19), 195131.
- 37 G. Román-Pérez and J. M. Soler, Efficient Implementation of a van der Waals Density Functional: Application to Double-Wall Carbon Nanotubes, *Phys. Rev. Lett.*, 2009, **103**(9), 096102.
- 38 K. Jiří, R. B. David and M. Angelos, Chemical accuracy for the van der Waals density functional, *J. Phys.: Condens. Matter*, 2010, **22**(2), 022201.
- 39 I. Hamada, van der Waals density functional made accurate, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **89**(12), 121103.
- 40 K. Lee, *et al.*, Higher-accuracy van der Waals density functional, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **82**(8), 081101.
- 41 Y. Ma, *et al.*, Graphene adhesion on MoS<sub>2</sub> monolayer: an ab initio study, *Nanoscale*, 2011, **3**(9), 3883–3887.
- 42 H. J. Monkhorst and J. D. Pack, Special points for Brillouin-zone integrations, *Phys. Rev. B: Solid State*, 1976, **13**(12), 5188–5192.
- 43 A. Bostwick, *et al.*, Quasiparticle dynamics in graphene, *Nat. Phys.*, 2006, **3**, 36.
- 44 G. Henkelman, A. Arnaldsson and H. Jónsson, A fast and robust algorithm for Bader decomposition of charge density, *Comput. Mater. Sci.*, 2006, **36**(3), 354–360.
- 45 C. Ataca, *et al.*, A Comparative Study of Lattice Dynamics of Three- and Two-Dimensional MoS<sub>2</sub>, *J. Phys. Chem. C*, 2011, **115**(33), 16354–16361.
- 46 Y. Liu and J. Wilcox, CO<sub>2</sub> Adsorption on Carbon Models of Organic Constituents of Gas Shale and Coal, *Environ. Sci. Technol.*, 2011, **45**(2), 809–814.
- 47 P. Cabrera-Sanfeliu, Adsorption and Reactivity of CO<sub>2</sub> on Defective Graphene Sheets, *J. Phys. Chem. A*, 2009, **113**(2), 493–498.
- 48 I. A. Baburin, *et al.*, Hydrogen adsorption by perforated graphene, *Int. J. Hydrogen Energy*, 2015, **40**(20), 6594–6599.
- 49 A. F. Ismail, K. Khulbe and T. Matsuura, Gas Separation Membranes: Polymeric and Inorganic, in *Membrane Technology and Applications*, Springer, 2015, p. 14.
- 50 B. Sachs, *et al.*, Doping mechanisms in graphene–MoS<sub>2</sub> hybrids, *Appl. Phys. Lett.*, 2013, **103**(25), 251607.
- 51 A. F. Holleman and E. Wiberg, *Inorganic Chemistry*, ed. N. Wiberg, Academic Press, San Diego, CA, USA, 2001.
- 52 J. Deng, *et al.*, Triggering the electrocatalytic hydrogen evolution activity of the inert two-dimensional MoS<sub>2</sub> surface via single-atom metal doping, *Energy Environ. Sci.*, 2015, **8**(5), 1594–1601.
- 53 T. F. Jaramillo, *et al.*, Identification of Active Edge Sites for Electrochemical H<sub>2</sub> Evolution from MoS<sub>2</sub> Nanocatalysts, *Science*, 2007, **317**(5834), 100–102.
- 54 J. Li, *et al.*, Enhanced CO<sub>2</sub> capture on graphene via N, S dual-doping, *Appl. Surf. Sci.*, 2017, **399**, 420–425.
- 55 S. Zhao, J. Xue and W. Kang, Gas adsorption on MoS<sub>2</sub> monolayer from first-principles calculations, *Chem. Phys. Lett.*, 2014, **595–596**, 35–42.
- 56 Z. Jijun, *et al.*, Gas molecule adsorption in carbon nanotubes and nanotube bundles, *Nanotechnology*, 2002, **13**(2), 195.
- 57 W.-L. Yim, *et al.*, Vibrational behavior of adsorbed CO<sub>2</sub> on single-walled carbon nanotubes, *J. Chem. Phys.*, 2004, **120**(11), 5377–5386.
- 58 CRC, *CRC Handbook of Chemistry and Physics*, CRC Press, Boca Raton, FL, 90th edn, 2009–2010.
- 59 C. Tayran, *et al.*, Structural and electronic properties of AB- and AA-stacking bilayer-graphene intercalated by Li, Na, Ca, B, Al, Si, Ge, Ag, and Au atoms, *Solid State Commun.*, 2016, **231–232**, 57–63.
- 60 S. Yang, *et al.*, Sodium adsorption and intercalation in bilayer graphene from density functional theory calculations, *Theor. Chem. Acc.*, 2016, **135**(7), 164.
- 61 C. Ataca and S. Ciraci, Functionalization of Single-Layer MoS<sub>2</sub> Honeycomb Structures, *J. Phys. Chem. C*, 2011, **115**(27), 13303–13311.
- 62 K. F. Mak, *et al.*, Atomically Thin  $\text{MoS}_2$ : A New Direct-Gap Semiconductor, *Phys. Rev. Lett.*, 2010, **105**(13), 136805.
- 63 A. M. Ukpong, First principles study of van der Waals heterobilayers, *Computational Condensed Matter*, 2015, **2**, 1–10.

